



Synthesis and Characterization of Doped CeO₂ as a Surrogate Nuclear Fuel

Alexander T. Nadermann^{1,2}, Adrianna E. Lupercio^{1,3}, Cayden Doyle^{1,3}, Riley C. Winters^{1,3}, Andrew T. Nelson⁴, Brian J. Jaques^{1,3}

1. Boise State University, College of Engineering, Boise, ID

2. Northwest Nazarene University, College of Natural and Applied Sciences, Nampa, ID

3. Center for Advanced Energy Studies, Idaho Falls, ID

4. Oak Ridge National Lab, Oak Ridge, TN



I. Background

Motivation for Research

- Nuclear energy continues to provide safe, productive energy, but can be improved.
- During the fission process, gasses are released into the fuel-cladding gap, lowering the thermal conductivity of the gap which leads to the reduction in efficiency and reliability of the fuel. [1]
- Increase in fuel grain size leads to an increase in the retention of fission gases.
- This study analyzes the impact of MnO₂ and TiO₂ additives on the grain size and microstructure of CeO₂ as a surrogate nuclear fuel for UO₂.

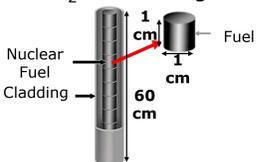


Figure 1. Schematic representation of a nuclear fuel and cladding of typical dimensions. [2]

Surrogate Nuclear Fuels

- CeO₂ can be used as a surrogate fuel for UO₂ due to similar thermodynamic properties and crystal structure (Figure 2). [3]
- Handling and processing highly radioactive materials produce many challenges.
- Surrogate fuels can resolve some of these challenges by:

- Increasing experimental timeliness
- Reducing cost
- Reducing radioactive exposure

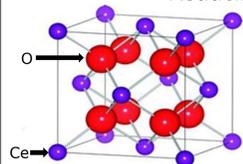


Figure 2. CeO₂ and UO₂ have the same fluorite crystal structure. [4]

II. Experimental

Materials Synthesis

- As-received CeO₂ was doped with MnO₂, and TiO₂ powders to obtain desired dopant concentrations.
- Each powder mixture was planetary ball milled at 250 rpm for 6 hours and high energy planetary ball milled (HEPBM) at 500 rpm for 1 hour to reduce the particle size and incorporate the dopants into the CeO₂ lattice.
- HEPBM powders were mixed with 0.45 wt% ethylene bis-stearamide (EBS) binder to improve pellet integrity prior to powders being pressed into pellets.
- Green pellets were pressed using a dual action die set at 150 MPa to obtain maximum theoretical density (TD).
- TiO₂-doped CeO₂ samples were sintered at 1600°C, while sintering temperatures ranged from 1200°C to 1550°C for MnO₂-doped CeO₂ samples.
- Samples were polished to 0.5 μm and thermally etched approximately 150°C below sintering temperature for characterization techniques.

Characterization Techniques

- X-Ray diffraction (XRD), with a lanthanum hexaboride (LaB₆) standard, was used for phase purity and dopant incorporation analysis.
- Optical microscopy and scanning electron microscopy (SEM) was used for pellet grain size and microstructure analysis.
- Energy dispersive spectroscopy (EDS) was used for a qualitative chemical analysis.
- Inductively coupled plasma mass spectroscopy (ICP-MS) was used for a quantitative chemical analysis.

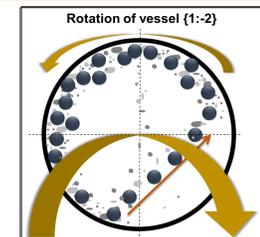


Figure 3. Schematic view of planetary ball mill motion.

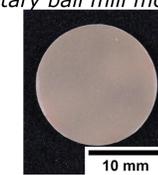


Figure 4. Sintered TiO₂-doped CeO₂ pellet.

III. Results

XRD Analysis of Doped Cerium Dioxide

▲ LaB₆ PDF 01-073-1669 ● CeO₂ PDF 03-065-2975 ■ TiO₂ PDF 01-088-1173

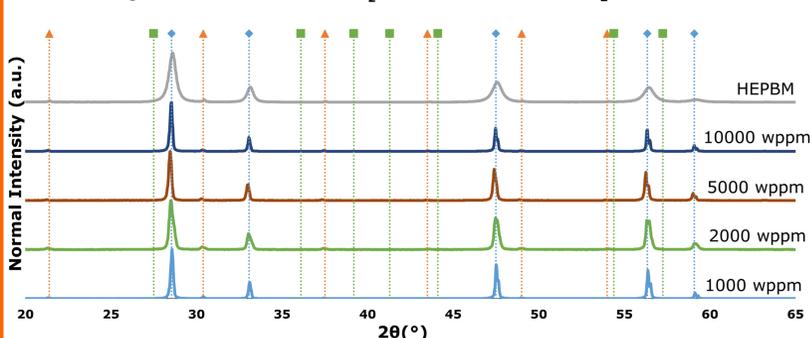


Figure 5. XRD of HEPBM and sintered TiO₂-doped CeO₂ indicates CeO₂ phase is maintained with no secondary TiO₂ peaks present. All HEPBM TiO₂-doped CeO₂ powders produced equivalent patterns and are combined on this plot.

▲ LaB₆ PDF 01-073-1669 ● CeO₂ PDF 03-065-2975 ■ MnO₂ PDF 01-071-0071

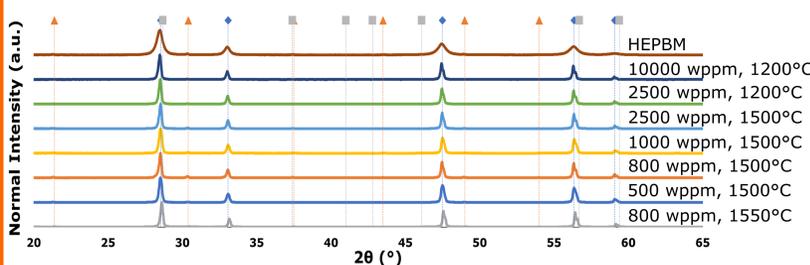


Figure 6. XRD of HEPBM and sintered MnO₂-doped CeO₂ indicates CeO₂ phase is maintained with no secondary MnO₂ peaks present. All HEPBM MnO₂-doped CeO₂ powders produced equivalent patterns and are combined on this plot. MnO₂ dopant concentrations and sintering temperatures are labeled accordingly.

Grain Size Analysis of Sintered Pellets

- Top surface and cross sections for each sample variant were imaged using an optical microscope and SEM, and the images were used to perform grain size analysis using ASTM E112-12. [5]
- Combining the top surface and cross section data results in a range of the average grain size for TiO₂-doped and MnO₂-doped CeO₂ (Figure 7).

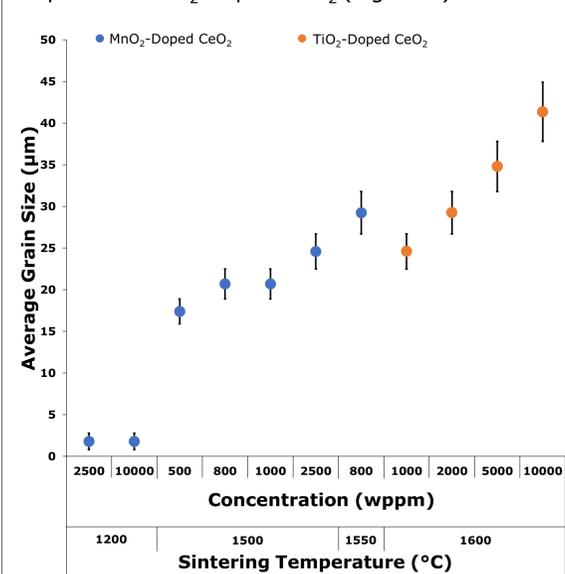


Figure 7. Average grain size range of MnO₂-doped and TiO₂-doped CeO₂.

Microstructural Analysis

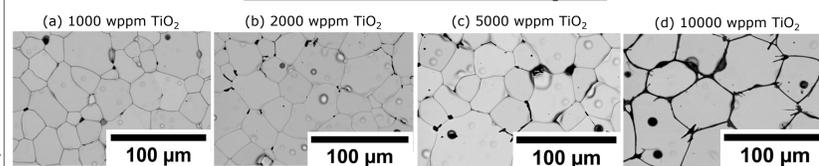


Figure 8. SEM images of TiO₂-doped CeO₂ samples show grain size, porosity and uniformity of samples. As dopant concentration is increased, sample grain size and porosity increase, reducing the density.

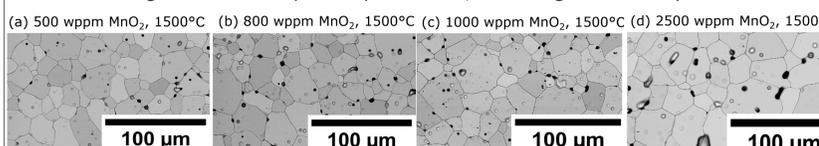


Figure 9. SEM images of MnO₂-doped CeO₂ samples show grain size, porosity and uniformity of samples. Sintering temperature is labeled for each sample. Samples sintered at 1500°C increased in porosity with dopant concentration (a-d).

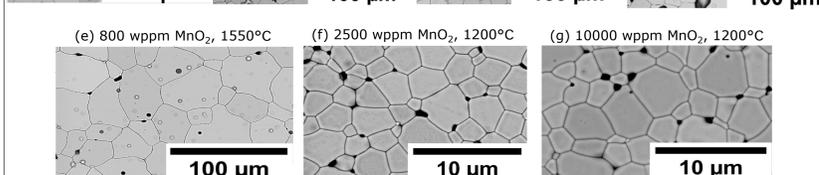


Figure 9. SEM images of MnO₂-doped CeO₂ samples show grain size, porosity and uniformity of samples. Sintering temperature is labeled for each sample. Samples sintered at 1500°C increased in porosity with dopant concentration (a-d).

Qualitative Chemical Analysis

- EDS performed on each sample variant is qualitative due to instrument detection limits (approximately 1 wt%).
- Trace amounts of Si were detected on samples and is likely due to the crucible used during sintering and thermal etching processes.
- EDS identified Ti⁺ in TiO₂-doped samples of concentrations ≥ 2000 wppm, while no Mn⁺ was detected in MnO₂-doped samples.

Quantitative Chemical Analysis

Table 1. Using Archimedes density for calculations, density values assume a single phase and stoichiometric CeO₂ with a reference density of 7.128 g/cm³. ICP-MS provides concentrations of HEPBM and sintered powders. [6]

Sample Identification	Sintering Temperature (°C)	Percent of Theoretical Density (%)	Target Cation Concentration (wppm)	Actual HEPBM Cation Concentration (wppm)	Actual Sintered Cation Concentration (wppm)
TiO ₂					
1000 wppm	1600	93 ± 0.5	600	364 ± 13%	334 ± 13%
2000 wppm		95 ± 0.5	1199	711 ± 13%	729 ± 13%
5000 wppm		93 ± 0.5	2998	1835 ± 13%	640 ± 13%
10000 wppm		90 ± 0.5	5995	3537 ± 13%	1043 ± 13%
MnO ₂					
2500 wppm	1200	96 ± 0.5	1580	1516 ± 14%	1355 ± 14%
10000 wppm		96 ± 0.5	6319	5441 ± 14%	5038 ± 14%
500 wppm	1500	97 ± 0.5	316	295 ± 14%	124 ± 14%
800 wppm		96 ± 0.5	506	447 ± 14%	179 ± 14%
1000 wppm		95 ± 0.5	632	241 ± 14%	203 ± 14%
2500 wppm		90 ± 0.5	1580	1516 ± 14%	311 ± 14%
800 wppm	1550	94 ± 0.5	506	447 ± 14%	171 ± 14%

IV. Discussion

- MnO₂-doped samples: 800 wppm MnO₂ samples sintered at 1550°C resulted in the largest grain size with a uniform microstructure and a %TD ≥ 94 ± 0.5 (Figure 9.e).
- TiO₂-doped samples: Increasing TiO₂ dopant concentration resulted in an increased grain size with decreasing %TD (Table 1), likely due to exceeding the solubility limit of TiO₂ in CeO₂ (Figure 8.d).
- For HEPBM powders, ICP-MS data indicates an average 40% and ≤ 15% error in target doping concentrations for TiO₂ and MnO₂ dopant concentrations, respectively.
- The doping process will be further refined.
- Further analysis will be performed on the samples to confirm the Si contaminant source and its impact.

References

- [1] J. Rest, et al., Fission gas release from UO₂ nuclear fuel: A review, *J. Nucl. Mater.* 513 (2019) 310–345.
- [2] O.O. Noah, et al., "Investigation of Natural Convection Heat-Transfer Phenomena in Packed Beds: Lead-Way Toward New Nuclear Fuel Design." *J. of Nucl. Engin. and Rad. Sci.* 1.4 (2015).
- [3] Kim, H. S., et al., Applicability of CeO₂ as a surrogate for PuO₂ in a MOX fuel development. *J. of Nucl. Mater.* 378.1 (2008) 98–104.
- [4] A. Younis, et al., "Cerium Oxide Nanostructures and their Applications," *Functionalized nanomaterials*, (2016) 53–68.
- [5] ASTM Standard, E112-12: Standard Test Methods for Determining Average Grain Size, ASTM Int. E112-12 (2013) 1–27. <https://doi.org/10.1520/E0112-12.1.4>
- [6] J.T. Dunwoody, and A.T. Nelson, "Cerium Dioxide Pellet Fabrication and Characterization," Los Alamos National Laboratory, (2017).

Acknowledgements

This work was partially funded through the U.S. Department of Energy in collaboration with the Idaho National Laboratory in the In-Pile Instrumentation Initiative. In addition, work was funded through an NEUP-IUP fellowship (DOE-FOA-0001487) and Oak Ridge National Laboratory (DE-ACOS-000R22725). Support was also provided by the National Science Foundation via the REU Site: Materials for Society at Boise State University (DMR 1658076 and 1950305).